



## The role and position of iron in $0.8\text{CaZrO}_3\text{-}0.2\text{CaFe}_2\text{O}_4$

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**Abstract.** The aim of the study was to characterize the  $0.8\text{CaZrO}_3\text{-}0.2\text{CaFe}_2\text{O}_4$  composite structure with particular emphasis on the role and position of iron in the function of sintering temperature. The paper presents the results of  $^{57}\text{Fe}$  Mössbauer effect at room temperature. It was found that the increase of sintering temperature causes an increase in the amount of incorporated iron ions in the  $\text{CaZrO}_3$ -crystal structure. Based on Mössbauer spectroscopy analysis, it was found that three different environments of  $\text{Fe}^{3+}$  ions were observed in the obtained materials. Two of them corresponded to  $\text{CaFe}_2\text{O}_4$  phase and one was associated with the substitution of  $\text{Zr}^{4+}$  by  $\text{Fe}^{3+}$  in the  $\text{CaZrO}_3$  structure.

**Key words:** calcium ferrite • calcium zirconate • Mössbauer spectroscopy • perovskite • spinel

### Introduction

$\text{CaZrO}_3$  is a member of perovskite, which has two polymorphs. At temperatures lower than  $1750^\circ\text{C}$ , an orthorhombic phase is stable, where  $\text{Ca}^{2+}$  ions are situated between slightly deformed  $\text{ZrO}_6$  octahedrals. At temperatures higher than  $1750^\circ\text{C}$ , cubic phase is stable [1, 2]. High purity  $\text{CaZrO}_3$  can be obtained by the arc melting technique. The fused  $\text{CaZrO}_3$ , with strong Zr-O covalent bonding and almost pure Ca-O ionic bonding, is poreless and has the density close to the theoretical one [3].

$\text{CaZrO}_3$ , which is the only chemical compound in the CaO-ZrO<sub>2</sub> binary system, is characterized by high melting point ( $2345^\circ\text{C}$ ), thermal shock resistance, high strength and corrosion resistance against alkali oxides and cement clinker. Furthermore,  $\text{CaZrO}_3$  has been widely investigated because of its electrical properties.  $\text{CaZrO}_3$  is a p-type semiconductor. Modified by some trivalent cations (e.g.  $\text{In}^{3+}$ ,  $\text{Ga}^{3+}$  and  $\text{Sc}^{3+}$ ) is a protonic conductor in hydrogen atmosphere. Small excess of zircona or calcia in  $\text{CaZrO}_3$  structure cause oxygen ion conductivity. The analogous situation occurs in the case of  $\text{Al}_2\text{O}_3$ ,  $\text{MgO}$ ,  $\text{Y}_2\text{O}_3$  substitution. Therefore,  $\text{CaZrO}_3$  is a material being used as, for example, mechanical filters, resonators, capacitors, and refractory materials [4–10].  $\text{CaZrO}_3$  relatively easily incorporates actinides into its crystal structure.  $\text{CaZrO}_3$  is considered as a host material for spent nuclear fuel immobilization [11].

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$\text{CaFe}_2\text{O}_4$  with spinel structure is stable up to  $1228^\circ\text{C}$  [12]. Crystal structure of  $\text{CaFe}_2\text{O}_4$  is similar to the one formed in perovskite compounds as in  $\text{CaZrO}_3$  [13]. Therefore, it is interesting to study the  $0.8\text{CaZrO}_3\text{-}0.2\text{CaFe}_2\text{O}_4$  composite, which is the combination of perovskite and spinel structure compounds. This work is focused on determining the position of the iron in the studied composite depending on the synthesis temperature.

## Experiment

The  $0.8\text{CaZrO}_3\text{-}0.2\text{CaFe}_2\text{O}_4$  composite was synthesized by solid-state reaction, starting from pure  $\text{CaCO}_3$ ,  $\text{ZrO}_2$ , and  $\text{Fe}_2\text{O}_3$  commercial powders. Synthesis of the material was achieved by a two-step heating process. The first step at  $1200^\circ\text{C}$  was calcination. The recovered material was grounded and pressed under 30 MPa before the second step heat treatment at  $1400^\circ\text{C}$  and  $1700^\circ\text{C}$  with the 10 h soaking time at each temperature.

The X-ray diffraction analysis was performed at a room temperature, using monochromatic  $\text{Cu-K}\alpha$  radiation (Philips Panalytical X'Pert-Pro MPD diffractometer) in order to determine the phase composition of the fired materials. The crystal structure parameters were obtained using the full-profile Rietveld method [14] implemented in the FullProf software package [15].

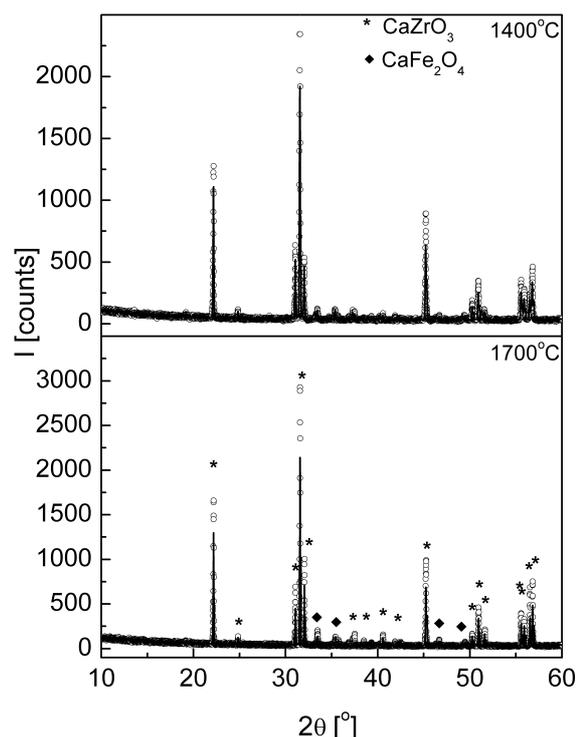
The  $^{57}\text{Fe}$  Mössbauer effect measurements were performed using the standard technique at room temperature in transmission mode, using a conventional constant-acceleration spectrometer and a 25 mCi  $^{57}\text{Co}$  source in Rh matrix. The velocity scale was calibrated using  $\alpha\text{-Fe}$  foil. Spectra were fitted to Lorentzian lines using the non-linear least square method.

## Results and discussion

Figure 1 shows X-ray diffraction (XRD) patterns and theoretical fits of the samples sintered at  $1400^\circ\text{C}$  and  $1700^\circ\text{C}$ . It was found that the main crystalline phases were  $\text{CaZrO}_3$  ( $Pcmm$ ) and  $\text{CaFe}_2\text{O}_4$  ( $Pnam$ ).

**Table 1.** Crystal structure parameters of  $\text{CaZrO}_3$  and  $\text{CaFe}_2\text{O}_4$  phases (a, b, c – unit cells parameters, V – unit cell volume, A – phase share percentage)

Parameters	Sintering temperature		JCPDS
	$1400^\circ\text{C}$	$1700^\circ\text{C}$	
$\text{CaZrO}_3$ ( $Pcmm$ )			
a [Å]	5.586(1)	5.582(1)	5.5912
b [Å]	8.011(1)	8.007(1)	8.0171
c [Å]	5.754(1)	5.754(1)	5.7616
V [Å <sup>3</sup> ]	257.54(1)	257.22(1)	258.26
A [%]	92.6(9)	88.0(9)	
$\text{CaFe}_2\text{O}_4$ ( $Pnam$ )			
a [Å]	10.756(9)	10.732(9)	10.705
b [Å]	9.262(4)	9.261(4)	9.23
c [Å]	3.027(4)	3.026(4)	3.024
V [Å <sup>3</sup> ]	301.56(7)	300.86(7)	298.79
A [%]	7.4(6)	12.0(6)	



**Fig. 1.** X-ray diffraction patterns and the Rietveld analysis of the samples sintered at  $1400^\circ\text{C}$  and  $1700^\circ\text{C}$ .

Parameters of both the crystalline phases are summarized in Table 1. With the sintering temperature increases the amount of the  $\text{CaFe}_2\text{O}_4$  phase at the cost of  $\text{CaZrO}_3$ , which could indicate the dissolution of  $\text{CaZrO}_3$  by  $\text{CaFe}_2\text{O}_4$  liquid phase. At both temperatures, a reduction of unit cell volume of  $\text{CaZrO}_3$  is observed. It may be caused by diffusion and incorporation of  $\text{Fe}^{3+}$  cations, which replaced bigger  $\text{Zr}^{4+}$  cations in the perovskite structure. An effective ionic radius of  $\text{Zr}^{4+}$  (0.72 Å) is considerably bigger than  $\text{Fe}^{3+}$  (0.55 Å) [16]. Thus if part of Zr is substituted by Fe, it should decrease the crystal structure parameters. The opposite effect is observed in the case of  $\text{CaFe}_2\text{O}_4$ . The diffusion and incorporation process are thermally activated. Therefore, at  $1700^\circ\text{C}$  in comparison to  $1400^\circ\text{C}$ , the lowest parameters of unit cell were obtained.

**Table 2.** The hyperfine interaction parameters (A – area of the subspectra, IS – isomer shift, QS – quadrupole split) of the sample prepared at 1400°C and 1700°C

Component	A [%]	IS [mm/s]	QS [mm/s]
1400°C			
1	63.7	0.338(4)	0.647(7)
2	28.5	0.364(1)	0.322(5)
3	7.8	0.437(5)	0.643(8)
1700°C			
1	54.4	0.313(16)	0.627(10)
2	26.1	0.353(1)	0.305(13)
3	19.5	0.429(9)	0.633(10)

The results of the Mössbauer spectroscopy measurements are shown in Fig. 2 and are summarized in Table 2. It was found that the obtained spectra are the result of three components, which correspond to the three different neighborhoods of Fe<sup>3+</sup> ions. In the case of components 1 and 2, the hyperfine interaction parameters correspond to pure CaFe<sub>2</sub>O<sub>4</sub> [17], where iron occurs in the octahedral and tetrahedral sites (approximately 2:1 contribution). In contrast, component 3 is derived from iron, which was incorporated into the CaZrO<sub>3</sub> structure and replaces zircon in octahedral sites. The substitution of Zr<sup>4+</sup>/Fe<sup>3+</sup> causes an imbalance of a cell charge. This is compensated by the presence of an oxygen octahedron distortion in the middle of which Fe<sup>3+</sup> cations are situated, or by the removal of one oxygen unit from an octahedron corner. Its coordination number declines from 6 to 5. This effect causes the appearance of the larger electric field gradient acting on the Fe<sup>3+</sup> ions, which leads to an increase of quadrupole splitting (QS) parameter. The increase of the sintering temperature causes the increase in

incorporation of Fe<sup>3+</sup> ions into the CaZrO<sub>3</sub> structure from approximately 8% to 20%.

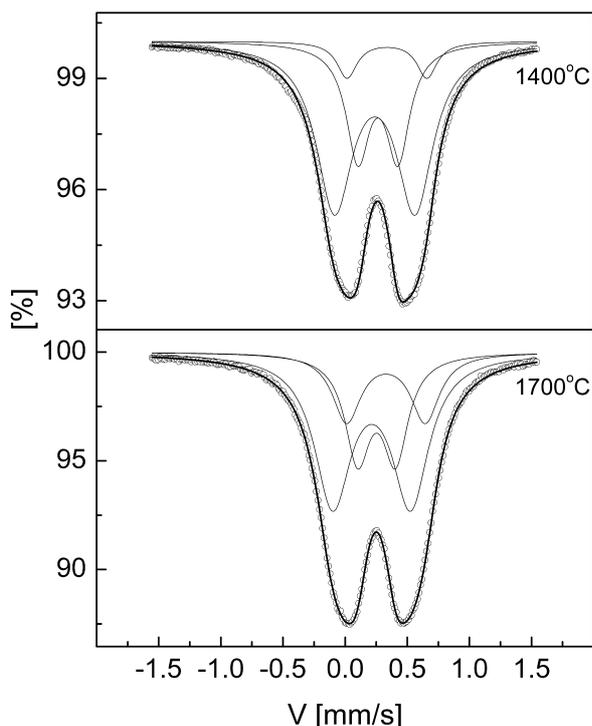
### Conclusions

The aim of the study is to characterize the 0.8CaZrO<sub>3</sub>-0.2CaFe<sub>2</sub>O<sub>4</sub> composite structure based on calcium zirconate with calcium ferrite in the matrix, with particular emphasis on the role and position of iron, as a function of temperature synthesis. Parameters of both the obtained crystalline phases: CaZrO<sub>3</sub> and CaFe<sub>2</sub>O<sub>4</sub> are slightly different, compared to the reference data, which may indicate the incorporation of iron in the CaZrO<sub>3</sub> crystal structure and zirconium in CaFe<sub>2</sub>O<sub>4</sub>. With the increase of the sintering temperature, a gradual dissolution of CaZrO<sub>3</sub> in the CaFe<sub>2</sub>O<sub>4</sub> liquid phase is observed. It was found that the sintering temperature increase causes an increase in the incorporation of iron ions into the crystal structure of calcium zirconate from approximately 8% to 20%. The study revealed that the CaZrO<sub>3</sub> structure is able to embody up to about 2 mol% of Fe<sup>3+</sup>.

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**Fig. 2.** <sup>57</sup>Fe Mössbauer effect measurements at room temperature of the sample sintered at 1400°C and 1700°C.

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