

# Effect of nanocrystals concentration on optical and luminescent properties of PVK:ZnSe nanocomposites

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This work presents a systematic study of the effect of ZnSe nanocrystals (NCs) concentration on the optical and luminescent properties of poly N-vinylcarbazole (PVK) polymer nanocomposites. The ZnSe nanocrystals were synthesized by a simple coprecipitation chemical route, while PVK:ZnSe nanocomposite films were fabricated using the spin coating technique. The samples were characterized by XRD, TEM, SEM, UV-Vis and fluorescence techniques. The X-ray diffraction and TEM studies confirmed the particle size, microstructure and spherical shape of the synthesized nanocrystals. The ZnSe nanocrystals in PVK caused a decrease in optical gap with increasing concentration of nanocrystals. The emission spectra exhibited augmentation in intensity up to 70 wt.% of nanoparticles while further addition resulted in a decrease in luminescence. The structure-property relationships obtained for the present system are important for developing low cost illumination devices.

Keywords: *nanocrystals; PVK:ZnSe nanocomposites; XRD; optical and fluorescence properties*

## 1. Introduction

Over the past years, organic light-emitting diodes (OLED) incorporating polymer matrices and inorganic nanocrystals have been developing very quickly for their potential application in flat panel displays. The use of semiconductor nanocrystals (NCs) received strong attention because of quantum confinement effects [1], imparting size tunable optical and electronic properties important for light-emitting diodes [2], photovoltaic devices [3], etc. The ZnSe semiconductor has a direct and wide-band gap (2.7 eV), and a large exciton binding energy (21 meV) leading to interesting physical properties. It is a good candidate for being used in light emitting devices and laser diodes. Many synthetic methods have been developed for preparing these materials, including photochemical method, sonochemical method [4], hydrothermal method [5], microwave heating process [6], microemulsion method [7], etc.

The fabrication of hybrid organic-inorganic materials favors the enhancement in the carrier injection and recombination processes which are very important for realization energy efficient and highly luminescent light emitting devices [8]. The synthesis of highly dispersed inorganic nanomaterials in transparent organic polymer matrices has a great advantage of easy fabrication and enhanced materials properties [9]. One of the most widely studied polymers is PVK which shows a good combination of electrical and optical properties making it a good candidate for optoelectronic applications such as light emitting diodes [10]. The resonant transfer of energy through formation of hybrid excitons is very sensitive to the external perturbations or varying in weight percentages in the heterostructure [11]. The most important mechanism is the exciton capture by NPs in a polymer matrix [12].

The capping of nanoparticles with polymers was found to increase the surface stability along with significant influence on the morphology and optical properties of nanocomposites [13]. However, there are only few reports

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on the influence of different concentrations on the characteristics of polymer based organic/inorganic hybrid nanocomposites. The nanocomposite PVK:ZnSe presents a promising area in the LEDs application as shown by Mastour et al. [14]. The increase in weight percentage leads to an increase in the fluorescence intensity. This result shows the importance of studying the optical absorption spectra of these nanocomposites in order to deduce the band gap energy.

In this paper, we report the effect of ZnSe nanocrystals concentration on the optical and luminescence properties of spin coated PVK:ZnSe nanocomposites. The enhancement in the luminescence with the concentration has been reported for the PVK:ZnSe nanocomposites.

## 2. Experimental

### 2.1. Materials

ZnSe nanoparticles were prepared by simple co-precipitation method in aqueous medium. All the chemicals: zinc chloride (Aldrich, Fisher), selenium powder (Loba Chemie), sodium sulfate (CDH), PVK (Aldrich) were of analytical grade and used as purchased without further purification.

### 2.2. Synthesis of ZnSe nanoparticles

The sodium selenosulfate was prepared by dissolving 5 mM selenium powder and 30 mM sodium sulfate in 100 mL distilled water to get sodium selenosulfate solution. The solution was continuously stirred for 8 h, kept overnight and finally filtered to get a clear solution of sodium selenosulfate for further reactions. For the synthesis of ZnSe nanoparticles, 0.1 M zinc chloride was dissolved in 100 mL distilled water and then sodium selenosulfate solution was added dropwise along with 1 mL of hydrazine hydrate as a stabilizer. Few drops of ammonia solution were added to get a transparent mixture. 1 M NaOH solution was further added into the mixtures to adjust the pH and to achieve precipitation of ZnSe nanoparticles [15]. Finally, the product was separated by centrifugation and washed with distilled water and ethanol several times.

The nanoparticles were finally dried in a hot air oven at 50 °C overnight to get a powdered sample.

### 2.3. PVK:ZnSe nanocomposite films

For the fabrication of PVK:ZnSe nanocomposites, PVK was dissolved in chloroform (2 mg/mL) and different concentrations of ZnSe nanoparticles (0 wt.%, 10 wt.%, 30 wt.%, 50 wt.%, 70 wt.% and 90 wt.%) were added to this solution and stirred at room temperature for 2 h. The solution was then coated on the well cleaned glass substrates (2000 rpm for 60 s) by using spin coating unit (Model: spin NXG-P1, Apex instruments, Kolkata). Finally, the samples were dried at 100 °C in a hot air oven.

### 2.4. Samples characterization

The synthesized ZnSe nanocrystals were characterized by X-ray diffractometer (Model: X'Pert PRO, PANalytical, UK) equipped with a goniometer (PW3050/60) using CuK $\alpha$  radiation ( $\lambda = 1.5406 \text{ \AA}$ ) in the  $2\theta$  range of 20° to 80°. The shape and microstructure of ZnSe nanocrystals were determined using transmission electron microscope (Model: JEM-2100, JEOL). Surface morphology of the nanocomposite films was analyzed using scanning electron microscope (Model: Quanta 200 F SEM, FEI, USA). The infrared spectrum of the nanoparticles was recorded on the FT-IR spectrophotometer (Model: RZX, PerkinElmer, USA). The absorption spectra of colloidal nanoparticles and nanocomposite films were recorded by UV-Vis spectrophotometer (Lambda-750, PerkinElmer). The emission spectra were recorded using a fluorescence spectrometer with a xenon flash lamp as excitation source (Model: Cary Eclipse, Agilent, USA).

## 3. Results and discussion

### 3.1. Characterization of ZnSe nanocrystals

Fig. 1a shows the X-ray diffraction pattern of pure ZnSe nanocrystals. The diffraction peaks at the  $2\theta$  angles of 29.57°, 45.23°, 56.08° and 61.42° correspond to the (1 0 1), (1 1 0), (0 0 4)

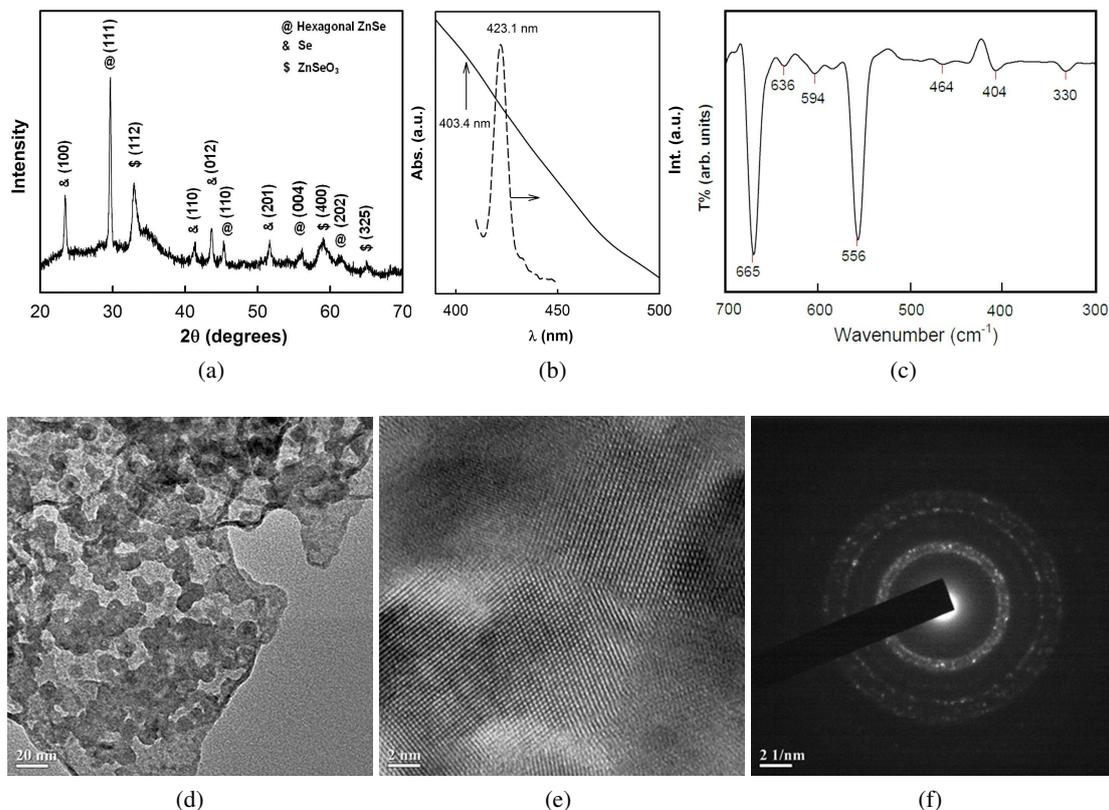


Fig. 1. (a) XRD pattern, (b) UV-Vis absorption and photoluminescence emission at 370 nm excitation energy, (c) FT-IR transmission spectrum, (d) TEM image (e) HRTEM image and (f) SAED pattern of the synthesized ZnSe nanoparticles.

and (2 0 2) planes of hexagonal structure of ZnSe (JCPDS Card No. 80-0008),  $23.43^\circ$ ,  $41.31^\circ$ ,  $43.54^\circ$  and  $51.58^\circ$  peaks can be assigned to (1 0 0), (1 1 0), (0 1 2) and (2 0 1) planes of Se (JCPDS Card No. 73-0465) and  $32.90^\circ$ ,  $58.94^\circ$ , and  $64.92^\circ$  correspond to (1 1 2), (4 0 0) and (3 2 5) planes of orthorhombic ZnSeO<sub>3</sub> (JCPDS Card No. 78-0446) [16, 17]. The formation of other phases may be due to the use of excess sodium selenosulfate along with the atmospheric air during the synthesis process. The crystallite size  $D$  was calculated using the Debye-Scherrer formula:

$$D = 0.9\lambda / \beta \cos\theta \quad (1)$$

where  $\lambda$  is the wavelength of radiation used,  $\beta$  is the full width at half maximum (FWHM) of diffraction peak, and  $\theta$  is the Bragg angle. The crystallite size calculated from the (1 0 1) plane of the ZnSe nanocrystals was found to be about

4.63 nm. The presence of a relatively sharp peak in a background of a wider peak suggests that both large and small grains were formed together.

Fig. 1b shows the optical absorption and luminescence spectra of the colloidal ZnSe nanoparticles [18]. It is observed that the absorption spectrum exhibits a peak at 403.4 nm (3.07 eV) and the blue shift from the bulk optical gap (2.70 eV) of the sample revealing the formation of ZnSe nanoparticles. The luminescence spectrum exhibits a strong peak at around 423.1 nm at the excitation energy 370 nm. Fig. 1c shows the IR transmission spectrum of the ZnSe nanocrystal in the frequency range of  $300 \text{ cm}^{-1}$  to  $700 \text{ cm}^{-1}$ . The main characteristic absorption peaks at  $556 \text{ cm}^{-1}$  and  $665 \text{ cm}^{-1}$  are observed and can be ascribed to stretching mode vibrations of Zn–Se bonds in ZnSe [19].

Fig. 1d shows the TEM micrograph of the colloidal ZnSe nanoparticles. This shows the particles are of spherical shape with an average size of 7.3 nm which is larger than that calculated from Scherrer method. Fig. 1e shows the HRTEM micrograph and reveals the interplanar spacing of 0.269 nm corresponding to (2 0 0) plane of ZnSe nanocrystals. Fig. 1f shows the selected area electron diffraction (SAED) pattern, exhibiting broad diffused rings corresponding to (1 1 1), (2 2 0) and (3 1 1) planes that are typical of nanosized ZnSe particles with a cubic structure [20]. Factors, such as supersaturation, nucleation and growth rates, colloidal stability, recrystallization and aging processes, have considerable effect on the particle size and microstructure [21].

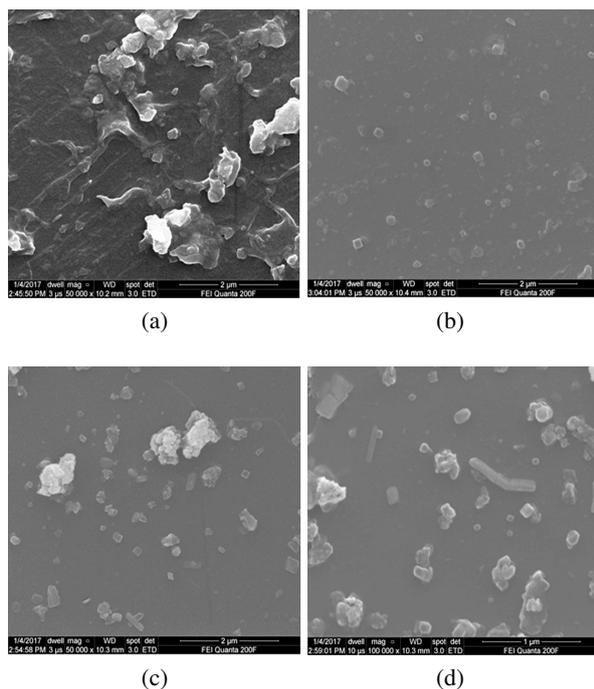


Fig. 2. FE-SEM images of spin coated (a) pure PVK, and (b) 30 %, (c) and (d) 70 % ZnSe nanoparticles in PVK:ZnSe films (various magnifications).

### 3.2. Characterization of PVK:ZnSe nanocomposites

The morphology of prepared nanoparticles was studied by FE-SEM. Fig. 2a to Fig. 2d show

the typical FE-SEM images of PVK polymer and PVK:ZnSe nanocomposites. Fig. 2a reveals the irregular shaped surface or porous structure resulting in higher surface roughness of the spin coated polymer films. The incorporation of nanoparticles resulted in an increase in surface smoothness with the emergence of the cubic shaped polymer clusters (the average dimensions are less than  $\sim 4$  nm) in the PVK:ZnSe nanocomposite film as shown in Fig. 2b [22]. However, the increase in ZnSe nanoparticle concentration favored the aggregation of the cube shaped structures as shown in Fig. 2c, while the enlarged image (Fig. 2d) reveals the cuboid shaped polymer structures on the surface of the films. These structures are typical of the polymer nanocomposites as reported in the literature [23].

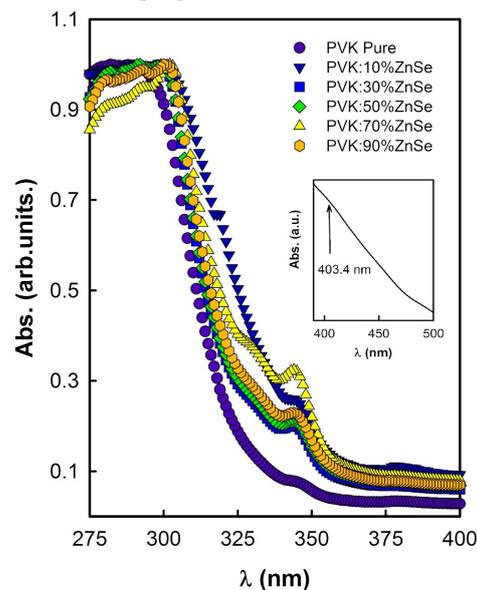


Fig. 3. Optical absorption spectra of spin coated PVK:ZnSe nanocomposite films (inset shows the absorption spectra for ZnSe nanoparticles).

Fig. 3 shows the optical absorption spectra of the PVK:ZnSe nanocomposites with different concentrations of ZnSe nanoparticles (inset shows the spectra for ZnSe nanoparticles). The sharp absorption edge (298 nm) accompanied with a small bump (345 nm) is attributed to  $n \rightarrow \pi^*$  optical transitions in the pure PVK films [24]. The incorporation of small amount of ZnSe nanoparticles (10 wt.%) resulted in a red shift

and decrease in steepness of absorption edge (304 nm) along with an increase in the absorbance of the bump in PVK:ZnSe nanocomposites. Similar values of the optical gap and red shift in the optical gap have been reported for the PVK:ZnSe composite films [25]. The anomalous variation in the absorption coefficient below the absorption edge can be ascribed to the combined effect of light scattering from the nanoparticles and their aggregated structures formed in the nanocomposite films. Further addition of nanoparticles resulted in an increase in absorption of the shoulder band along with anomalous variation in the absorption edge for the nanocomposites. These results suggest that the incorporation of ZnSe nanoparticles in PVK favored the enhancement in the intensity for  $n \rightarrow \pi^*$  optical transitions characteristic of the change in the stacking of polymeric units in the PVK:ZnSe nanocomposites.

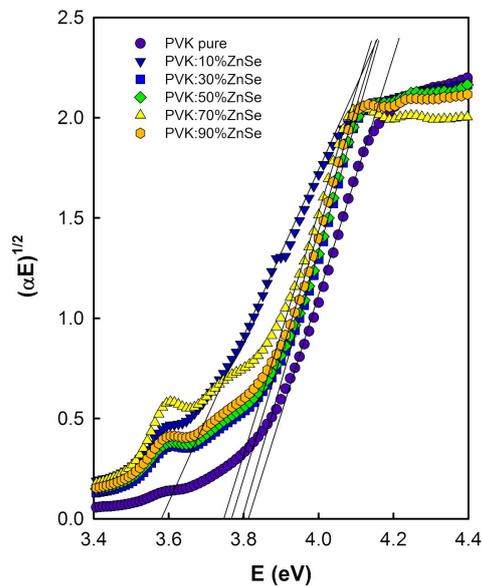


Fig. 4. Plot of  $(\alpha E)^{1/2}$  vs. photon energy  $h\nu$  for PVK:ZnSe nanocomposite films.

The absorption coefficient in the high absorption region can be related to the incident photon energy by [22]:

$$\alpha E = B(h\nu - E_o)^2 \quad (2)$$

where B is the quality factor of which the inverse  $B^{-1}$  represents the width of band tailing,  $h\nu$  is

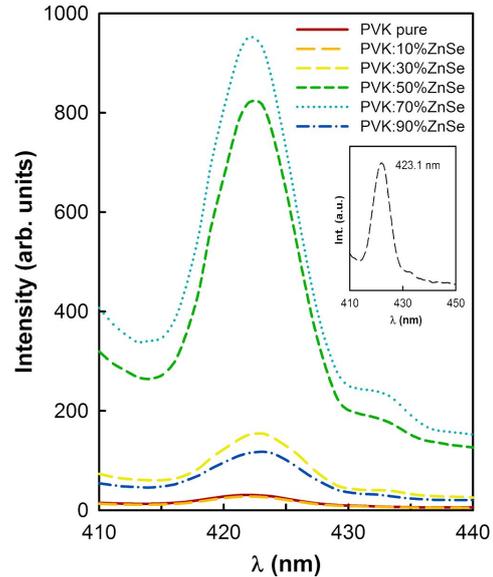


Fig. 5. Emission spectra of PVK:ZnSe nanocomposite films at 400 nm excitation energy (inset shows the emission spectra of ZnSe nanoparticles at 370 nm excitation).

the photon energy and  $E_o$  is the indirect optical gap. Fig. 4 shows the Tauc plot, i.e.  $(\alpha E)^{1/2}$  versus photon energy  $h\nu$  for the PVK:ZnSe nanocomposite films with different concentrations of nanoparticles. The optical gap of spin coated PVK film is 3.81 eV which is in a good agreement with the values reported in the literature. The band gap decreases to 3.59 eV with the incorporation of small concentration of ZnSe nanoparticles (10 %) in the PVK matrix. This change can be ascribed to large structural changes in the polymer network with nanoparticle addition. Further increase in the nanoparticle concentration resulted in an anomalous variation in the optical gap of the PVK:ZnSe nanocomposite films. Different types of modification of the polymer network, cluster elimination and their further formation were observed with changing nanoparticle concentration.

Fig. 5 shows the photoluminescence spectra of pure PVK and PVK:ZnSe nanocomposites with different concentrations of ZnSe nanoparticles (0 % to 90 %) at 400 nm excitation (the inset shows the luminescence spectra of ZnSe nanoparticles at 370 nm excitation). The spectra exhibit

two strong emission peaks centered at 420 nm and 423 nm, accompanied with a shoulder peak at 432 nm and attributed to the triplet-singlet transitions in pure spin coated PVK films [26]. It has been observed that the addition of nanoparticles (NP - x (x = 10 % to 90 %)) resulted in an enhancement in emission peak at 423 nm as well as in the 432 nm band (pure PVK) of the hybrid nanocomposites. The luminescence peaks are assigned to PVK as the excitation energy is lower than that used for the excitation of ZnSe nanoparticles. No new emission peaks have been observed in the luminescence spectra of this system, however, the new peaks related to nanoparticles in Cu<sub>2</sub>Se/PVK system were reported by Govindraju et al. [27]. The luminescence has been found to increase up to 70 wt.% of ZnSe nanoparticles addition in the nanocomposites, while further addition of the nanoparticles resulted in some decrease in the luminescence intensity. The enhancement in the intensity can be ascribed to the formation of good quality nanocomposite blends and surface passivation effect of PVK on the ZnSe nanoparticles in these hybrid materials. However, the decrease in the intensity may be due to the formation of polymer clusters or aggregates in the polymer nanocomposite. Similar results have been reported earlier for such systems [24].

#### 4. Conclusions

The effect of concentration of ZnSe nanocrystals on the optical and fluorescence properties of spin coated PVK nanocomposite films has been studied. The analysis of structure and optical properties confirmed the formation of spherical shaped ZnSe nanocrystals in the quantum confinement regime. The optical gap was found to decrease from 3.81 eV to 3.59 eV with the addition of 10 % ZnSe nanoparticles while anomalous variation in the optical properties has been observed for PVK:ZnSe nanocomposite films. The optical spectra exhibited a strong emission centered at 423 nm along with a shoulder peak at 432 nm for pure spin coated PVK films. The initial increase in concentration of ZnSe nanocrystals (up to 70 %) was found to increase the luminescence of PVK but it was found

to decrease with further addition of the nanocrystals. The charge transfer process and the aggregation of nanocrystals in the nanocomposites at higher concentrations have been used to explain the variation of fluorescence intensity of the synthesized nanocomposites. The obtained results are very important for designing new organic/inorganic hybrid materials for low cost illumination devices.

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