

Ab-initio studies of thermal and superconducting properties of HfX_2 alloys ($\text{X} = \text{Tc}, \text{Re}, \text{and Os}$)

V.S. SATHYAKUMARI*, S. SANKAR, K. MAHALAKSHMI

Department of Physics, Madras Institute of Technology Campus, Anna University, Chennai – 600044, India

A systematic study of thermal properties such as the Debye temperature, specific heat coefficient, Grüneisen constant, electron-phonon coupling constant and transition temperature have been carried out using the results of electronic band structure and related characteristics, for hafnium superconducting alloys, namely, HfTc_2 , HfRe_2 and HfOs_2 . Computation of the electronic band structure and associated properties has been carried out using the tight-binding-linear-muffin-tin-orbital (TBLMTO) method within atomic sphere approximation (ASA). The calculated values have been compared with the available results of literature data.

Keywords: *Debye temperature; specific heat coefficient; Grüneisen constant; electron-phonon coupling constant; transition temperature*

© Wrocław University of Technology.

1. Introduction

Hafnium forms a number of intermetallic binary alloys [1, 2] of the general formula HfX_2 . Several are hexagonal C14 Laves structure alloys. Some intermetallic compounds of Hf and the transition metals Fe, Co, Pd and Pt have been investigated as hydrogen-storage materials because of their capability to form hydrides with high hydrogen to metal ratios at room temperature [3]. Hafnium is primarily used in the control and safety mechanisms of nuclear reactors because of its high cross-section for neutron absorption and its high corrosion resistance [4]. Hafnium cladding of nuclear fuel rods is expected to be an important element in the design of future advanced reactors [5]. Hafnium alloys are also used in medical implants and devices, due to their biocompatibility and corrosion resistance [6]. Hafnium is added to aluminum-magnesium-scandium alloys, widely used in aerospace applications, to increase their strength during high temperature thermo mechanical processing [7]. The hafnium dirhenide compound was first reported by Comp-

ton and Matthias [8]. Krikorin et al. [9] determined the lattice parameters for the analyzed composition $\text{HfRe}_{2.02}$. The phase diagram for the hafnium-rhenium was published by Taylor, Kagle and Doyle [10]. They found that the HfRe_2 phase occurs over a range of homogeneity with the lattice parameters varying. In a similar study Savitskii et al. [11] also reported a range of homogeneity but listed the variation of lattice parameters. The superconducting transition temperature for the HfRe_2 has been variously reported as 4.80 K by Compton and Matthias [8] and 5.61 K by Blaugher, Taylor and Hulm [12] suggesting Tc changes with composition. The lattices parameters of the HfTc_2 compound have been given by Darby et al. [13]. The richness of phenomena and the properties observed in these materials have motivated the study on the properties of their fundamental electronic structures. Hence, in order to have such a comprehensive understanding of the basic electronic properties of these materials, we have carried out the electronic band structure studies using the well-known and versatile TBLMTO method and the results of electronic, ground state, thermal and superconducting properties of HfTc_2 , HfRe_2 and HfOs_2 are presented here. The results have been analyzed and compared with the literature data.

*E-mail: sathyakumari2003@gmail.com

2. Computational method

The hexagonal C14 Laves structure alloys HfTc_2 , HfRe_2 and HfOs_2 are interesting due to their thermal and superconducting properties [14, 19]. Space groups of HfTc_2 , HfRe_2 and HfOs_2 are the same as $\text{P6}_3/\text{mmc}$ (no. 194). The atomic position in HfTc_2 , HfRe_2 and HfOs_2 are: Hf1: 1/3, 2/3, 0.5629 and Tc₁ or Re₁ or Os₁: 0.1697, 0.3394, 1/4 and Tc₂ or Re₂ or Os₂: 0, 0, 0. The electronic configurations of elements in these materials are Hf: $[\text{Xe}] 4f^{14}5d^26s^2$; Tc: $[\text{Kr}] 4d^55s^1$ Re: $[\text{Xe}] 4f^{14}5d^56s^2$ and Os: $[\text{Xe}] 4f^{14}5d^66s^2$. The band structure calculations for these intermetallic compounds were carried out by using the TB-LMTO method within the ASA [15, 16]. The density of states (DOS) was calculated by the method of tetrahedron [17]. All the muffin-tin radii and number of k-points were varied to ensure total energy convergence. The ground-state properties were calculated for the lattice parameters that corresponded to the minimum of the total energy.

3. Results and discussion

All the electronic properties such as the DOS and energy bands were calculated for the equilibrium lattice parameters. For each case, equilibrium lattice parameters were obtained by minimizing the electronic total energy (E_T) with respect to the cell volume. The plots of total energy as a function of volume for HfTc_2 , HfRe_2 and HfOs_2 are shown in Fig. 1 (a – c) respectively. The cell volumes have been varied by varying the a-parameter and maintaining c/a ratio constant. The curve was obtained by fitting the E_T -V data to Murnaghan's equation of state [18]:

$$E(V) = V \frac{B_0}{B'_0} \left[\frac{(V_0/V)^{B'_0}}{B'_0 - 1} + 1 \right] + \text{const.} \quad (1)$$

where B_0 is bulk modulus and B'_0 is the pressure derivative of B_0 . The equilibrium lattice parameters were obtained by minimizing the total energy by varying the cell volumes of HfTc_2 , HfRe_2 and HfOs_2 alloys. The minimum energy value corresponds to the volume of 1320.8 \AA^3 for HfTc_2 and the corresponding lattice constant is 5.1475 \AA . In

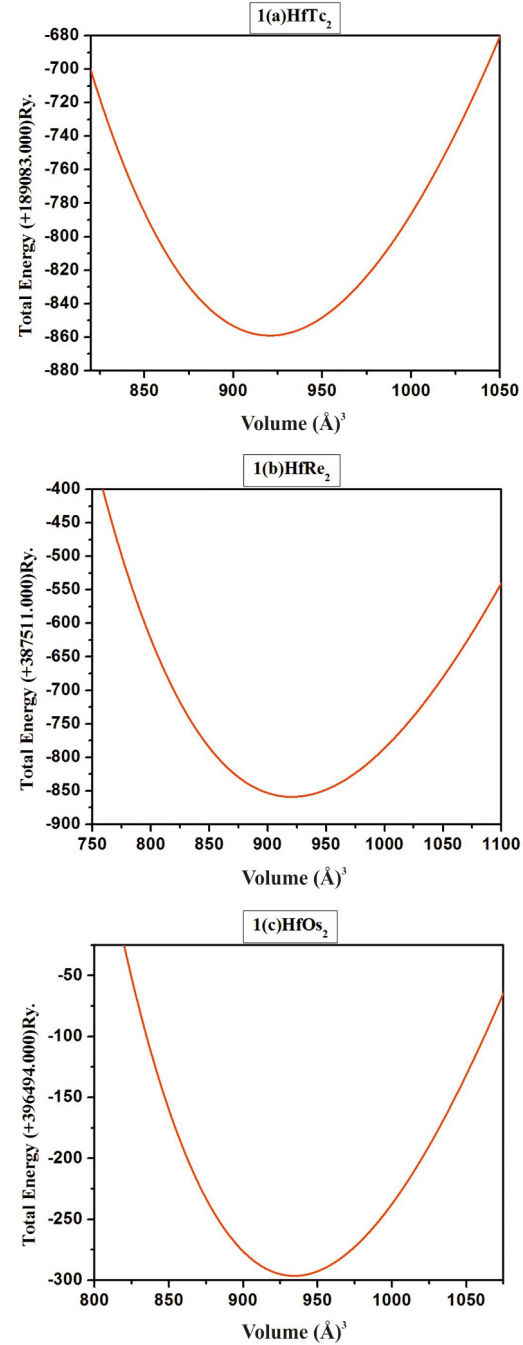


Fig. 1. Variation of total energy with cell volume for (a) HfTc_2 , (b) HfRe_2 and (c) HfOs_2 .

a similar manner to volume optimization, the lattice constants were calculated; they are: 5.2187 \AA and 5.1652 \AA for HfRe_2 and HfOs_2 , respectively. The variations of lattice constants are in compliance with the variation of the size of atoms in their

alloys. A comparison with the experimental lattice parameters of HfTc_2 , HfRe_2 and HfOs_2 [14, 19] is given in Table 1.

Table 1. Comparison of the calculated lattice parameters with the experimental results.

Alloys	Lattice parameter Å	
	Present	Experimental
HfTc_2	$a = 5.1475$	$a = 5.2001$ [14]
	$c = 8.5305$	$c = 8.6175$
HfRe_2	$a = 5.2187$	$a = 5.2540$ [14]
	$c = 8.5423$	$c = 8.6000$
HfOs_2	$a = 5.1652$	$a = 5.2000$ [19]
	$c = 8.4350$	$c = 8.4920$

Transition metals have large d-electron shells and are characterized by high binding energy [20]. The electronic band structures of HfTc_2 , HfRe_2 and HfOs_2 , along the high symmetry directions of the Brillouin zone are displayed in Fig. 2 (a – c). The Fermi level is set to 0 Ry. The bands are intense around the Fermi energy and are mainly derived from the d-states of Tc_1 , Re_1 , and Os_1 atoms of HfTc_2 , HfRe_2 and HfOs_2 , respectively. These are indicating the domination of d-bands. It is clearly evident in all the three materials that these states dominate the conduction in these materials. At higher energies there is a hybridization of the d-states of the hafnium. Just above the Fermi level the bands arise mostly from the p-states Tc_1 , Re_1 , and Os_1 atoms of HfTc_2 , HfRe_2 and HfOs_2 , respectively. They are also very strongly hybridized. The Fermi energies of HfTc_2 , HfRe_2 and HfOs_2 are: 0.796676Ry, 0.83377Ry and 0.841479Ry, respectively. A relatively larger value of E_F for HfOs_2 is attributed to the more occupied p- and d-bands, relative to the other two compounds.

The DOS of all the three alloys, namely HfTc_2 , HfRe_2 and HfOs_2 , are presented in Fig. 3 (a – c). The DOS at E_F for HfTc_2 , HfRe_2 and HfOs_2 , are presented in Table 2. The high DOS around E_F is evidently due to the strong piling up of p- and d-bands in all the three alloys and accounts for the structural stability. It is clearly evident that d-bands of Tc_1 , Re_1 , and Os_1 atoms of HfTc_2 , HfRe_2

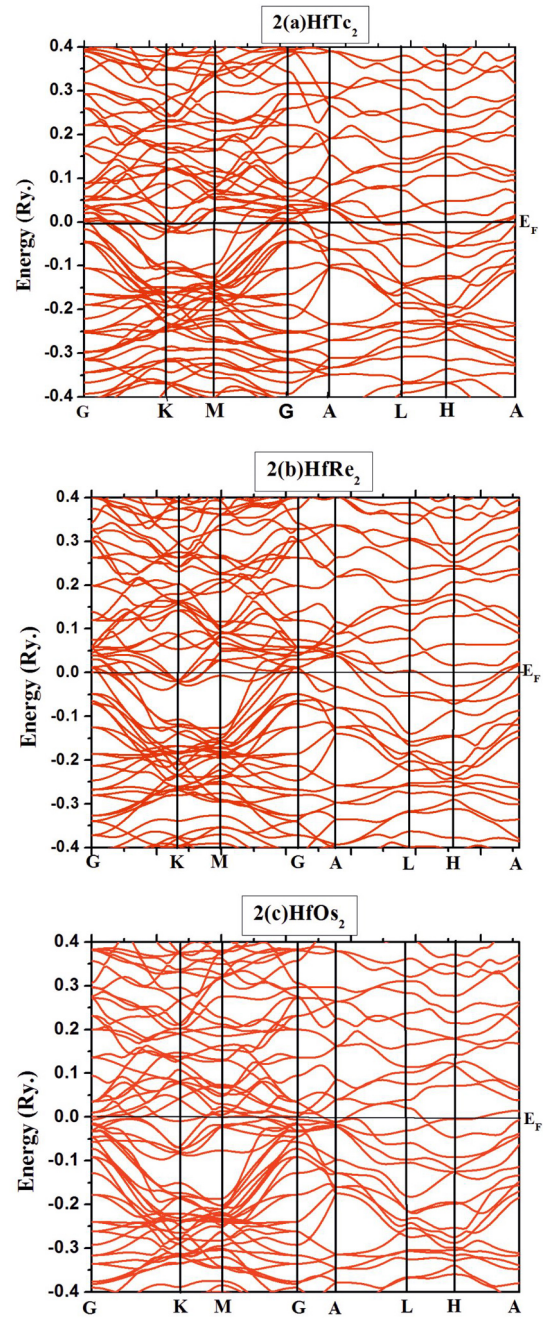


Fig. 2. Energy band structure for (a) HfTc_2 , (b) HfRe_2 and (c) HfOs_2 .

and HfOs_2 , respectively contribute predominantly to the Fermi level for all the materials. The p-bands of Tc_1 , Re_1 , and Os_1 atoms of HfTc_2 , HfRe_2 and HfOs_2 , respectively also contribute a little to E_F but there is no contribution from p-bands of Tc_2 ,

Table 2. Electronic density of states at the Fermi energy for HfTc_2 , HfRe_2 and HfOs_2 for s-, p-, d- and f-bands of various atomic elements.

Bands	DOS at E_F ((states/Ry)/atom)											
	HfTc_2				HfRe_2				HfOs_2			
	Hf ₁	Tc ₁	Tc ₂	Total	Hf ₁	Re ₁	Re ₂	Total	Hf ₁	Os ₁	Os ₂	Total
s	1.565	0.596	0.677	2.838	1.338	0.616	0.301	2.255	2.757	0.704	0.186	3.647
p	2.939	7.020	1.955	11.914	3.974	6.502	2.562	13.038	3.748	13.643	1.759	19.150
d	38.222	68.828	17.748	124.798	44.618	62.998	15.925	123.541	31.749	45.126	14.625	91.500
f	2.216	1.584	1.123	4.923	2.619	1.926	1.350	5.895	2.324	1.189	0.962	4.475

Re_2 , and Os_2 atoms of HfTc_2 , HfRe_2 and HfOs_2 to E_F . The DOS at the Fermi level is intense due to d-states. The DOS at E_F is found to decrease drastically for HfOs_2 in comparison to HfTc_2 and HfRe_2 . This is due to the increase in atomic number of X_2 elements. The Fermi energy moves towards the valley point next to the d-band peak in the DOS for HfOs_2 . For HfTc_2 and HfRe_2 , the valley point is near the Fermi level lying around the maximum region of the DOS plots and hence, the DOS at E_F for these alloys are high.

The Debye temperature (Θ_D) and Grüneisen constant (γ_G) are important parameters, related to the thermal properties of materials. To calculate the thermal properties of a vibrating Debye lattice we have used the Debye-Grüneisen model [21]. In the present work, the Debye temperature is calculated using the equation 2:

$$\Theta_D = 67.48 \sqrt{\frac{r_0 B_0}{M}} \quad (2)$$

where r is the Wigner-Seitz radius in a.u., $r = r_0$ (at equilibrium) B_0 is the bulk modulus in kbar and M is the average atomic weight. The Grüneisen constant (γ_G) is related to the Debye temperature (Θ_D) and the volume of the unit cell (V) by the relation [22]:

$$\gamma_G = \frac{\partial \ln \Theta_D}{\partial \ln V} \quad (3)$$

The calculated values of Θ_D and γ_G are shown in Table 3. The Debye temperature is found to agree reasonably well with the existing experimental reports [19]. The electronic specific heat coefficient

Table 3. Equilibrium Wigner-Seitz radii (r_0), bulk moduli (B_0), Debye temperature (Θ_D) and Grüneisen constant (γ_G) for HfTc_2 , HfRe_2 and HfOs_2 .

Alloys	r_0 (a.u.)	B_0 (GPa)	Θ_D (K)		γ_G
			Present	Exp.	Present
HfTc_2	2.97297	356.29	359	–	1.05
HfRe_2	3.00171	395.42	313	–	1.20
HfOs_2	2.96860	434.29	324	330 [19]	0.78

(γ) is related to the density of states $N(E_F)$ and the electron phonon mass enhancement factor (λ) by the expression

$$\gamma = (1 + \lambda) \frac{1}{3} \pi^2 k_B^2 N(E_F) \quad (4)$$

where k_B is the Boltzmann constant. The electron-phonon coupling constant (λ) can be estimated using the electronic band structure results from the relation:

$$\lambda = \frac{N(E_F) \langle I^2 \rangle}{M \langle \omega^2 \rangle} \quad (5)$$

where I^2 is the electron-phonon interaction matrix element averaged over the Fermi surface [23], M is the molecular mass, and $\langle \omega^2 \rangle$ is the average phonon frequency square given by [24]:

$$\langle \omega^2 \rangle = \frac{1}{2} \Theta_D^2 \quad (6)$$

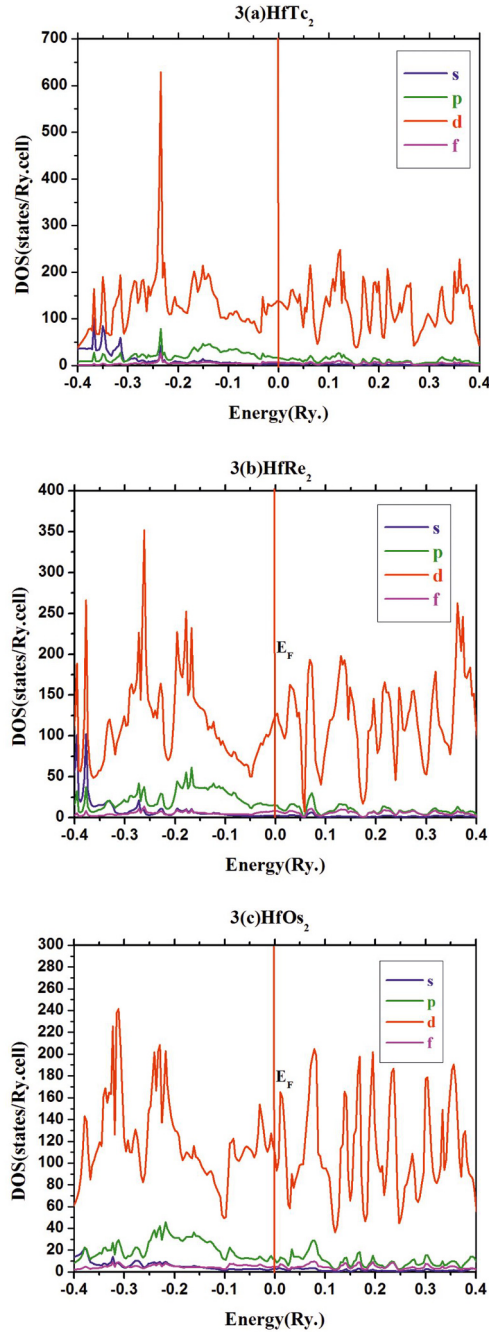


Fig. 3. Density of states for (a) HfTc₂, (b) HfRe₂ and (c) HfOs₂.

with Θ_D in energy units. The quantity $\langle I^2 \rangle$ is determined by the relation:

$$\langle I^2 \rangle = 2 \sum_l \frac{(l+1)}{(2l+1)(2l+3)} M_{l,l+1}^2 \frac{N_l(E_F) N_{l+1}(E_F)}{N(E_F) N(E_F)} \quad (7)$$

where $M_{l,l+1}$ is the electron-phonon matrix element which can be expressed in terms of the logarithmic derivatives (D_l):

$$M_{l,l+1} = -\phi_l \phi_{l+1} [(D_l(E_F) - l)(D_{l+1}(E_F) + l + 2) + (E_F - V(S))S^2] \quad (8)$$

where $V(S)$ is the one electron potential and ϕ_l is the sphere boundary amplitude of the l partial wave function evaluated at E_F . N_l in equation 7 is the partial density of states function for the angular momentum quantum number l . The values of the electronic specific heat coefficient (γ) for HfTc₂, HfRe₂ and HfOs₂ are given in Table 4. For HfOs₂, $\gamma^{exp} = 2.84 \text{ mJmol}^{-1}\text{K}^{-2}$ [19]. No such data have been found in the literature for other two compounds for comparison. The superconducting transition temperature (T_C), which is obtained by using the procedure is detailed below. The superconducting transition temperature has been calculated using McMillan's formula [25], which is:

$$T_C = \frac{\langle \Theta_D \rangle}{1.45} \exp \left[\frac{-1.04(1+\lambda)}{\lambda - \mu^*(1+0.62\lambda)} \right] \quad (9)$$

where λ is the electron-phonon coupling constant which can be estimated as detailed above, and μ^* is the electron-electron interaction constant. The μ^* can be obtained from the empirical relation given by Bennemann and Garland [26]:

$$\mu^* = \frac{0.26N(E_F)}{1 + N(E_F)} \quad (10)$$

It is evident from Table 4 that among the three compounds HfRe₂ has the highest T_C value. The T_C values estimated by the present work for HfTc₂, HfRe₂ and HfOs₂ agree well with the experimental values reported earlier [14, 19]. No such results have been available for λ , Θ_D and γ of HfTc₂ and HfRe₂ in the literature for comparison. As the atomic number of X in HfX₂ (X = Tc, Re, and Os) increases, the values of λ , γ and superconducting transition temperature T_C , are found to decrease.

4. Conclusions

The electronic, thermal and superconducting properties of hexagonal C14 Laves structure alloys

Table 4. Electron-phonon interaction constant (λ), electronic specific heat coefficient (γ), and superconducting transition temperature (T_C) for HfTc₂, HfRe₂ and HfOs₂.

Alloys	λ		γ (mJmol ⁻¹ K ⁻²)		T_C (K)	
	Present	Exp.	Present	Exp.	Present	Exp.
HfTc ₂	0.6	–	2.26	–	4.50	5.60 [14]
HfRe ₂	0.51	–	2.09	–	5.10	5.20 [14] 4.80 [27]
HfOs ₂	0.43	0.49 [19]	1.75	2.84 [19]	1.86	1.85 [19] 2.69 [27]

of HfTc₂, HfRe₂ and HfOs₂ superconductors have been studied using the self-consistent tight binding linear muffin-tin orbital method. The ground state lattice parameters and primitive cell volume have been obtained by fitting the electronic total energy with the Murnaghan equation of state and are in good agreement with the experimental results. The electronic band structures show a strong hybridization between p- and d-bands and are also narrowly piled up around the Fermi energy for all the three materials and it is also evident in the density of states results. Furthermore, it has been found that the band structure and related properties at the Fermi energy are dominated by the d-bands in general. The density of states at Fermi energy is found to be strongly influenced by the d-bands due to X-atoms (X = Tc, Re, Os) in all the three compounds. The total d-band density of states at Fermi energy is found to decrease with the increasing atomic number of X-atoms in the compound, and therefore, the electron-phonon interaction parameter, electronic specific heat, and superconducting transition temperature have all been found to follow the same trend of variation. There are however, some discrepancies between the present theoretical estimates of T_C and the corresponding experimental values. The probable source of this difference is mainly attributed to the empirical estimation of the pseudo potential parameter (μ^*) for the electron-electron interactions using the Bennemann-Garland formula (the correct estimation of it requires more involved renormalization procedure by taking into account spin fluctuation as suggested earlier by Picket et al. [28]). Thus, within the approximations involved in the calculations, the present results are found to agree reasonably well with the available experimental results.

References

- [1] LEVY O., HART G.L.W., CURTAROLO S., *Acta Mater.*, 58 (8) (2010), 2887.
- [2] DARBY JR. J.B., LAM D.J., NORTON L.J., DOWNEY J.W., *J. Alloy Compd.*, 4 (1962) 558.
- [3] BAUDRY A., BOYER P., FERREIRA L.P., HARRIS S.W., MIRAGLIA S., PONTONNIER L., *J. Phys.-Condens. Mat.*, 4 (1992), 5025.
- [4] *ASM Handbook, vol. 13B*, in: Cramer S.D., Covino B.S. (Eds.), *Corrosion: Materials, Crystal Structure*, ASM International, Materials Park, Ohio, 2005.
- [5] WALLENIUS J., WESTIEN D., *Ann. Nucl. Energy*, 35 (2006), 60.
- [6] DAVIDSON J.A., *Titanium molybdenum hafnium alloys for medical implants and devices*, US Patent No. 5954724, (September 21, 1999).
- [7] FERNADES M.T., *Aluminum-magnesium-scandium alloys with hafnium*, WO/2001/012868, World Intellectual Property Organisation, 2001.
- [8] COMPTON V.B., MATTHIAS B.T., *Acta Crystallogr.*, 12 (1959), 651.
- [9] KRIKORIAN N.H., WITTEMAN W.G., BOWMAN M.G., *J. Phys. Chem.*, 64 (1960), 1517.
- [10] TAYLOR A., KAGLE B.J., DOYLE N.J., *J. Alloy Compd.*, 5 (1963), 26.
- [11] SAVITSKII E.M., TYLKINA M.A., TSYGANIVA I.A., GLADYSHEVSKII E.I., MULYAVA M.P., *Russ. J. Inorg. Chem.*, 7 (1962), 831.
- [12] BLAUGHER R.D., TAYLOR A., HULM J.K., *IBM J. Res. Dev.*, 6 (1962), 116.
- [13] DARBY J.B., LAM D.J., NORTON L.J., DOWNEY J.W., *J. Alloy Compd.*, 4 (1962), 558.
- [14] GIORGI A.L., SZKLARZ E.G., *J. Alloy Compd.*, 22 (1970), 246.
- [15] ANDERSEN O.K., *Phys. Rev. B*, 12 (1975), 3060.
- [16] ANDERSEN O.K., JEPSEN O., *Phys. Rev. Lett.*, 53 (1984), 2571.
- [17] JEPSEN O., ANDERSEN O.K., *Solid State Commun.*, 9 (1971), 1763.
- [18] MURNAGHAN F.D., *P. Natl. Acad. Sci. USA*, 30 (1944), 5390.
- [19] KUENTZLER R., WATERSTRAT R.M., *J. Alloy Compd.*, 125 (1986), 261.
- [20] KITTEL CH., *Introduction to Solid State Physics*, 7th ed., John Wiley and Sons, INC., U.K., 1995.
- [21] DEBYE P., *Ann. Phys.-New York*, 39 (1912), 786.

- [22] SHASHIKALA H.D., MOHANRAO P.V., MURTHY K.S.N., SURYANARAYANA S.V., *J. Phys.C.*, 20 (1987), 2063.
- [23] SKRIVER H.L., MERTIG I., *Phys. Rev. B*, 32 (1985), 4431.
- [24] ASHWANIKUMAR D., OJHA P., *J. Supercond. Nov. Magn.*, 24 (2011), 1385.
- [25] MCMILLAN W.L., *Phys. Rev.*, 167 (1965), 331.
- [26] BENNEMANN K.H., GARLAND J.W., IN: DOUGLASS D.H. (ED.), *Superconductivity in d- and f-bands metals*, American Institute of Physics, New York, 1971.
- [27] BERGER L.I., ROBERTS B.W., *Properties of Superconductors CRC Hand Book of Chemistry and Physics*, 90th Ed., CRC Press, Inc., Boca Raton, Florida, USA, 2010.
- [28] PICTET O., JARLBORG T. PETER M., *J. Phys. F-Met. Phys.*, 17 (1987), 221.

Received 2013-12-13

Accepted 2014-04-10