

INVESTIGATION OF SILICON CARBIDE POLYTYPES
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Polytypes of colourless and coloured single crystals of silicon carbide (SiC) grown on SiC substrates by chemical vapour deposition are studied using Raman spectroscopy supplemented by scanning electron microscopy (SEM) and X-ray diffraction (XRD) analyses. The SEM analysis of the defect stacking faults, inclusions of defects and their distribution has shown that they correlate with the peak positions of the obtained Raman spectra and with the XRD data on the crystal structure.

Keywords: *Silicon carbide (SiC), polytypes, Raman spectroscopy, X-ray diffraction (XRD).*

1. INTRODUCTION

Silicon carbide (SiC) is a highly attractive material for fabrication of microelectronic and optoelectronic devices due to its wide band gap, high thermal conductivity, excellent thermal and chemical stability as well as its resistance to radiation damage and electrical breakdown [1]. As known, silicon carbide crystallizes in many polytypes of so-called polymorphs, their total number exceeding 200 [2]. The list of the most common polytypes includes 3C, 2H, 4H, 6H, 8H, 9R, 10H, 14H, 15R, 19R, 20H, 21H, and 24R, where (C), (H) and (R) are the basic crystallographic categories: cubic, hexagonal and rhombohedral. In the cubic category, zinc-blend structures are labelled 3C-SiC or β -SiC. The hexagonal category is represented by n H-SiC and the rhombohedral – by n R-SiC polytypes, which are generally referred to as α -SiC, n Si-C bilayers consisting of C- and Si-layers stacked in a unit cell [2].

The most common forms are 4H, 6H (known as hexagonal α -SiC types), and the cubic 3C-SiC type (β -SiC) [2]. Among diversified polytypes, the 3C-SiC variety possesses unique properties, including a high electron mobility up to 1000 cm²/V·s and a consequent high saturation drift velocity. Therefore, reproducible growth of SiC single crystals on silicon or silicon carbide substrate is of high

importance for the semiconductor industry. The limitations of SiC technologies stem from structural defects (point defects, line (or 2D) defects, stacking faults), which adversely affect the operation of semiconductor devices. Single crystals of a high-purity cubic polytype are usually colourless or yellow, while of 6H polytypes – grey or green. The colour of a crystal depends on the type of impurity and its concentration in the crystal. By varying the nitrogen content in a 6H-SiC polytype crystal it is possible to change its colour from a translucent light green to the opaque dark green.

In the present work, the Raman spectroscopy – a fast and contact-free method with easy sample preparation – supplemented by X-ray diffraction (XRD) and scanning electron microscopy (SEM) techniques was applied for characterization of SiC polytypes [3,4] and impurities in SiC [5]. Under study are colourless and coloured silicon carbide single crystals grown on SiC substrates by chemical vapour deposition using the Raman spectroscopy in combination with the SEM and XRD analyses.

2. EXPERIMENTAL

All investigated SiC samples were grown by direct synthesis from silicon and carbon vapours in deep vacuum in a quasi-closed system of high-purity graphite. The samples were deposited onto SiC substrates. The grown SiC single crystals differed in colour and transparency depending on the conditions during the growth process. Our experiments were carried out with colourless and coloured (grey, yellow, blue and green) SiC single crystals. Investigation of their physical properties has shown that the crystal colour is determined primarily by the polytype, and only then by the influence of impurities entering the crystal during growth.

The morphology of the as-synthesized SiC, its defective structure and the density of defects were studied by means of optical microscopy and using a scanning electron microscope (SEM, *TESCAN Mira* \LMU FEG). The chemical composition of samples was analyzed with the energy-dispersive X-ray spectroscopy (EDX, *Oxford System* embedded in the SEM). The SiC crystal structure was analyzed using Cu K α radiation (a *PANalytical X'Pert PRO* X-ray diffraction system).

The micro-Raman scattering measurements were carried out at room temperature (RT) in the backscattering geometry using an *Advantage Raman NIR* spectrometer (resolution 3 cm⁻¹, 785 nm laser) and a *Spex Ramalog Raman* spectrometer (resolution 1 cm⁻¹, 532 nm and 633 nm lasers).

3. RESULTS AND DISCUSSION

Sample growing

For growing the SiC single crystals an original technology and a specially designed device were applied. The technology [6] is based on the method of direct

SiC synthesis from silicon and carbon in different phases: silicon – as liquid or vapour, and carbon (graphite) – as solid (vaporizes at the temperature of 2000 °C and vacuum of 10^{-5} - 10^{-6} mBar).

The synthesis of SiC proceeds in the region above the melt due to high concentration of silicon and carbon vapours and high temperature. Once the SiC compound is formed, in the presence of a temperature gradient these vapours move to the cooler zone and deposit on the substrate where a crystal is growing.

To find the optimal conditions of Si evaporation and SiC synthesis as well as the optimal temperature gradients, the crystal growth rates and the design of equipment, in the experiments we varied the temperatures of Si evaporation, SiC synthesis and crystal growth. Since the most valuable SiC crystals – both coloured and colourless – are those without defects, the greatest challenge for us was to grow SiC crystals with minimum defects. The defects in SiC crystals that are optically visible are the following: splices of different polytypes, voids, cracks, inclusions, etc. Most of these defects arise owing to the presence of impurities in the growing crystal.

X-ray diffraction

The X-ray diffraction picture of the obtained grey single crystals shows a good compliance of the most intense peaks with the hexagonal 6H-SiC polytype. This was revealed on the 002 and 004 peaks corresponding to the lattice parameters $a = 3.0763 \text{ \AA}$ and $c = 15.080 \text{ \AA}$. The crystal surface is oriented parallel to the (002) plane. Simultaneously, aside from the diffraction maxima typical of the (002) plane (and of the equivalent ones) with the intensities characteristic of a single crystal's diffraction maxima, separate low-intensity peaks are observed, which is indicative of small crystallite inclusions of another orientation.

In the green SiC single crystal with defects the intense 002 and 004 peaks are observed that correspond to the hexagonal triclinic structure of 2H-SiC with lattice parameters $a = 0.3076 \text{ nm}$ and $c = 0.5048 \text{ nm}$ and cubic structure 3C-SiC with $a = 0.4359 \text{ nm}$. The crystal surface is oriented parallel to the (111) plane.

XRD patterns of the green SiC single crystal with defects are shown in Fig.1. The peaks in Fig.1 are identified as belonging to the hexagonal polytypes K_{a1} and K_{a2} 2H-SiC and the cubic polytypes K_{a1} and K_{a2} 3C-SiC.

The 3C-SiC and 2H-SiC polytypes are stable in the temperature range from 1300°C to 1600°C, while the 2H-SiC polytype transforms to the 3C-SiC polytype at temperatures 1400 -1800°C.

Figure 2 shows the SEM image of the green SiC single crystal with defects.

To determine the chemical composition of the green single crystal with defects, the EDX measurements were taken using SEM – separately for the areas with and without defects. The EDX analysis has shown that all the samples contain only silicon and carbon. The results evidence that the atomic percentage of carbon is higher in the defect-free areas (C – 54-56% and Si – 44-46%) than in those with defects (C–33-35% and Si – 65-67%).

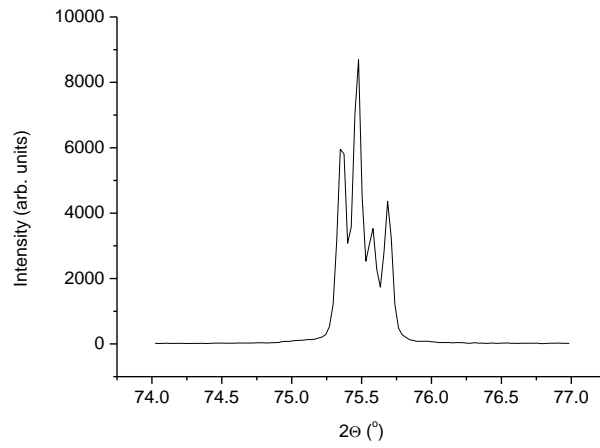


Fig.1. XRD pattern of the green SiC single crystal with defects.

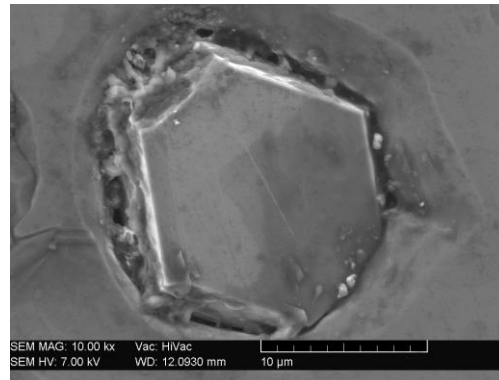


Fig.2. SEM image of the SiC green single crystal with defects.

Raman Spectroscopy

The results of Raman analysis performed for two samples: the colourless SiC single crystal and the grey SiC single crystal are presented in Fig.3 and Table 1.

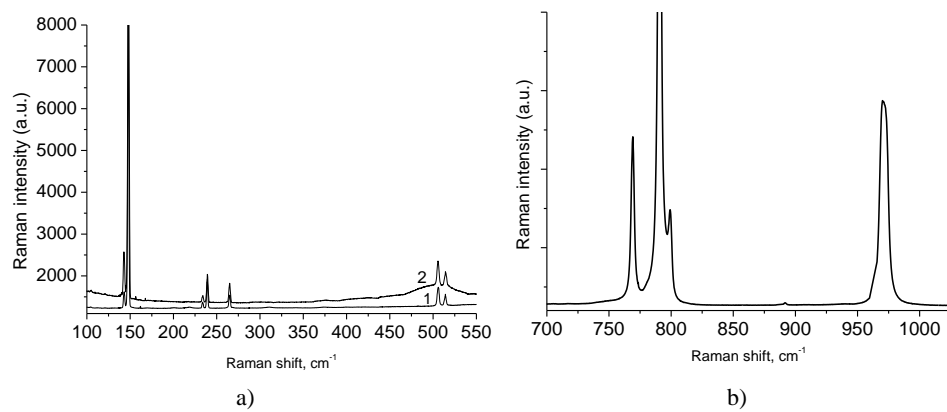


Fig.3. RT micro-Raman spectra of:
a) 1 – colourless SiC single crystal, 2 – grey SiC single crystal (Raman shift 100-550 cm^{-1});
b) colourless SiC single crystal (Raman shift 700-1000 cm^{-1}).

Raman spectra of colourless and grey single crystals in Fig. 3 and their frequencies (Raman shifts) in Table 1 are indicative of polytype 6H-SiC, which is in agreement with the data reported in [7]. All 6H-SiC samples possess the hexagonal wurtzite structure, which is confirmed by the X-ray diffraction measurements.

Table 1

Raman shifts of the peaks shown in Fig.3a,b with reference to [7]

Colourless single crystal Raman shift, cm^{-1}	Grey single crystal Raman shift, cm^{-1}	6H-SiC peak [7] Raman shift, cm^{-1}	Phonon Mode [7]
143, 148	143, 148	146.0, 150.5	E2 planar acoustic
234, 238	234, 239	235.0, 240.0	E1 planar acoustic
265	265	266.0	E2 planar acoustic
506, 514	504, 515	505.0, 513.5	A1 axial acoustic
769, 791	769, 791	767.5, 788.0	E2 planar optic
799	798	796.0	E1(TO)
892	892	888.5	A1 axial optic
970	970	966.5	A1(LO)

Figure 4 shows micro-Raman spectra of the green SiC single crystal for the areas with and without defects.

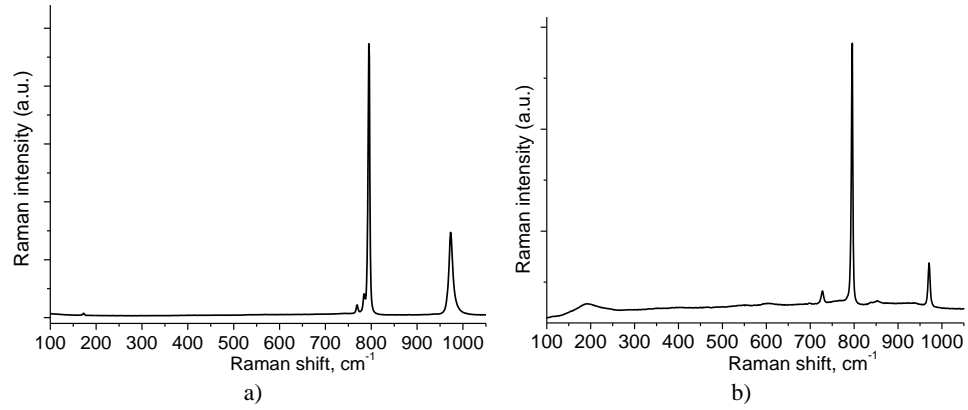


Fig.4. Micro-Raman spectra of the green SiC single crystal:
(a) defect-free area;
(b) area with defects.

The results of measuring the Raman spectrum of green SiC single crystal (Fig.4a) evidence that the defect-free area of the sample presents a cubic polytype 3C-SiC of the 795 cm^{-1} transverse optical (TO) mode and 973 cm^{-1} longitudinal

optical (LO) mode [8], and a 15R-SiC polytype with small crystallite inclusions (770 cm^{-1} (TO) and 172 cm^{-1} transverse acoustic (TA) mode). Figure 4b shows a typical Raman spectrum for the area of green SiC single crystal where the growth defects are observed. The Raman peaks of about 796 and 971 cm^{-1} correspond to the peaks of a rarer polytype 2H-SiC. The broad peak at 200 cm^{-1} can be attributed to the TA mode, the peaks at 728 and 786 cm^{-1} – to the TO mode, and the peak at 971 cm^{-1} – to the LO mode [9].

4. CONCLUSIONS

Raman spectra of different silicon carbide polytypes grown using our original method are helpful in detection of the defects in single crystals and of the presence of different phases (polytypes) which may arise in the volume of the main single crystal due to minor changes in the temperature or in other parameters during the process of crystal growth.

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SILĪCIJA KARBĪDA POLITIPU PĒTĪJUMI AR RAMANA SPEKTROSKOPIJU

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Kopsavilkums

Dažādu fāzu silīcija karbīda (SiC) monokristāli, kas audzēti uz SiC pamatnēm ar ķīmiskās nogulsņēšanas metodi no gāzveida fāzes, tika pētīti, izmantojot Ramana spektroskopiju, skenējošo elektronu mikroskopiju (SEM) un rentgenstaru difrakciju (XRD). Ar SEM palīdzību tika identificēti kristalogrāfiskās struktūras apgabali un ieslēgumi, ir pierādīts, ka tie korelē ar Ramana spektru pīķu pozīcijām, un XRD datiem par kristālisko struktūru.

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